5

10

15

20

# A PROCESS FOR IMPROVING THE HYDROLYSIS RESISTANCE OF A URETHANE ELASTOMER

# **BACKGROUND OF THE INVENTION**

#### 1. Field of the Invention

This invention refers to a process for improving the hydrolysis resistance of elastomers based on urethane, a polyester polyol for implementing this process, the urethane elastomer obtained by this process, and its use.

2. Description of the Related Art

Urethane elastomers are known to be prepared by causing a prepolymer, which is the reaction product of a polyester polyol or polyether polyol resin and an organic diisocyanate, to react with a hydroxylated compound consisting of at least one polyester polyol or polyether polyol in the presence of a catalyst and possibly a foaming agent and a surfactant.

In order to prepare the polyester polyol both on the prepolymer side as well as the hydroxylated compound side, carboxylic diacids such as adipic acid are commonly used.

The urethane elastomers produced in this way, based essentially on polyesters, possess good physical properties. However, for certain applications they are excessively sensitive to water. This is true when these elastomers are intended to form certain soles of shoes; it is also true when they are used to produce mechanical parts requiring a good resistance to hydrolysis. Urethane elastomers based on polyethers are not sensitive to hydrolysis but generally possess poorer physical properties.

In order to improve the hydrolysis resistance of polyester-based urethane elastomers, anti-hydrolysis additives of the polycarbodiimide type, such as that sold by Bayer under the trade name Stabaxol, are known to be added. Still, such additives are relatively costly and do not systematically achieve an improvement in the resistance to

5

hydrolysis. Indeed, in two out of three cases, the addition of such additives was found not to provide the expected improvement.

According to another approach, patent EP 0,156,665 proposes the use of dimerized fatty acids to prepare the polyester polyols used in the formulation of either the hydroxylated compound or the prepolymer. This solution provides improved results, but they are still insufficient, and furthermore the final properties are affected. The use of branched polyols has also been proposed for the preparation of polyester polyols. In this case the use of such polyols considerably increases the cost of the urethane elastomers and worsens their properties.

#### SUMMARY OF THE INVENTION

15

10

The purpose of this invention is to propose a process to improve the resistance to hydrolysis of a urethane elastomer that is effective, inexpensive, and compatible with the machines customarily used, while still ensuring that said elastomer will have good mechanical properties.

To this end, this invention proposes a process for improving the hydrolysis

20

25

30

resistance of a urethane elastomer, with said elastomer being prepared by the reaction of a hydroxylated compound with a prepolymer in the presence of a catalyst, and optionally a foaming agent and a surfactant, with said hydroxylated compound consisting of at least one polyester polyol resin, with said prepolymer being prepared by the reaction of at least a second polyester polyol resin and/or at least one polyether polyol resin and at least one polyisocyanate, with the polyisocyanate being in molar excess with respect to the second polyester polyol resin and/or polyether polyol resin, and with both the first and second polyester polyol resins being prepared by the reaction of at least one polyacid with at least one polyol, wherein at least the first polyester polyol resin is prepared by the reaction of at least one aliphatic dicarboxylic acid comprising between 8 and 12 carbon atoms and ortho-phthalic acid or its corresponding

5

anhydride, with at least one polyol.

## BRIEF DESCRIPTION OF THE DRAWINGS

Figs. 1-4 show in graphic form the hydrolytic resistance results of the elastomers of the invention and various comparative elastomers.

## DESCRIPTION OF THE PREFERRED EMBODIMENTS

10

The filing company has discovered, quite surprisingly, that by using a particular polyester polyol obtained by selecting specific acids, the hydrolysis resistance of the resulting urethane elastomers is considerably improved. According to the invention, the polyester polyol is prepared by the reaction of ortho-phthalic acid or its corresponding anhydride and at least one aliphatic dicarboxylic acid comprising between 8 and 12 carbon atoms, with at least one polyol. The process of the invention makes it possible to produce elastomers having an improved hydrolysis resistance under the usual implementation conditions, while still having good mechanical properties. In particular, the resulting elastomers quite surprisingly possess very good impact absorption properties.

20

15

Although the polyester polyol can also be used to prepare the prepolymer, use of the polyester polyol according to the invention solely for the preparation of the hydroxylated compound makes it possible to achieve excellent hydrolysis resistance results.

25

According to one feature, in order to prepare at least the first polyester polyol resin, the molar ratio of the ortho-phthalic acid or anhydride to the aliphatic dicarboxylic acid is between 30:70 and 60:40.

Advantageously, the aliphatic dicarboxylic acid used to prepare at least the first polyester resin is sebacic acid (C10).

Advantageously, the polyol used to prepare at least the first polyester polyol

resin is chosen from among the group consisting of monoethylene glycol (MEG), diethylene glycol (DEG), butanediol (BD), and their mixtures. The polyol used is preferably diethylene glycol (DEG).

10

15

20

25

When the polyester polyol of the invention is used solely to prepare the hydroxylated compound, said prepolymer is prepared by the reaction of at least one polyisocyanate and at least one polyether polyol resin and/or a second polyester polyol resin preferably formed from a polyacid chosen from among the group consisting of succinic acid (C4), glutaric acid (C5), adipic acid (C6) and their mixtures, and a polyol chosen from among the group consisting of monoethylene glycol (MEG), diethylene glycol (DEG), butanediol (BD) and their mixtures, with said polyol possibly being used in a mixture with a small proportion of another polyol chosen from among the group consisting of trimethylolpropane, glycerol, or pentaerythritol.

Another purpose of this invention is to propose a polyester polyol for the implementation of the process defined previously. The polyester of this invention is characterized by the fact that it is made by the reaction of at least one aliphatic dicarboxylic acid with C8-C12, preferably sebacic acid, and ortho-phthalic acid or its corresponding anhydride, with at least one polyol.

According to one feature, the molar ratio of the ortho-phthalic acid or anhydride to the aliphatic dicarboxylic acid is between 30:70 and 60:40.

The polyol employed may be chosen from among the group consisting of monoethylene glycol (MEG), diethylene glycol (DEG), butanediol (BD) and their mixtures. The polyol used is preferably diethylene glycol.

The purpose of this invention is also urethane elastomers obtained by the process of the invention and the use of these elastomers in order to make shoe soles.

The process of the invention for making hydrolysis-resistant urethane

10

15

20

25

elastomers can be described in greater detail as follows.

The polyester polyol resins of the invention are obtained by known means by causing at least one aliphatic dicarboxylic diacid with C8-C12 and ortho-phthalic acid or its corresponding phthalic acid anhydride to react with at least one polyol using a conventional esterification catalyst. A reaction vessel is loaded with the mixture of the aliphatic dicarboxylic diacid(s) of ortho-phthalic acid or the corresponding anhydride and the polyol(s) used. The phthalic/aliphatic molar ratio is between 20:80 and 70:30, and is preferably between 30:70 and 60:40. We heat and agitate in the atmosphere of an inert gas which may be nitrogen. A catalyst is optionally introduced. The water formed in the course of the reaction distills around 150°C for 1 to 4 hours and the temperature gradually rises. The temperature is kept at a value of between 160 and 250°C until obtaining an acid number (AN) of less than 2, preferably less than 0.5 mg KOH/g, and a hydroxyl number (OHN, number of OH groups) of between 25 and 230 mg KOH/g. The total reaction time is 15 to 30 hours.

Among the aliphatic carboxylic diacids having C8 to C12 that can be used, that is, suberic acid (C8), azelaic acid (C9), sebacic acid (C10), undecanedioic acid (C11), and dodecanedioic (C12), sebacic acid is preferred.

The esterification catalysts that can be used may be chosen from among the organic titanates, the alkylaryl and alkylsulfonic acids, the phosphoric acids, or the tin salts. The catalysts may advantageously be present in the reaction mixture in a concentration of between 0.004% and 0.020% by weight in relation to the weight of the reaction mixture.

This polyester polyol of the invention is part of the composition of the hydroxylated compound and may possibly be used to prepare the prepolymer.

In order to prepare the prepolymer, at least one polyester polyol resin and/or

polyether polyol resin and at least one isocyanate having a functionality of at least 2, known for the preparation of polyurethanes, whether they are aliphatic, cycloaliphatic, arylic, arylaliphatic, or heterocyclic, are caused to react together. Generally preferred are the polyisocyanates that are easily available on the market, such as 2,4- and 2,6-toluylene diisocyanate (TDI) and their mixtures, or pure 4,4'-methylene-diphenyl-diisocyanate (MDI) and its derivatives and their mixtures, in a polycarbodiimide-modified grade such as, for example, the product marketed by Bayer under the trade name "Desmodur CD" or the product marketed by Dow under the trade name "Isonate 143L." The isocyanate portion is in molar excess with respect to the polyester resin so as to obtain a prepolymer with isocyanate ends that can react with the hydroxylated compounds to obtain a urethane elastomer.

The prepolymer may be prepared from a polyester polyol consisting of the polyester polyol of the invention or a previously known polyester polyol, and/or from any previously known polyether polyol. Consequently, any prepolymer available on the market may be used for the implementation of the invention. As an example, existing polyester polyols that are used are prepared by condensation of at least one aliphatic dicarboxylic acid with at least one polyol. Among the polyols that can be used advantageously, let us mention as an example, monoethylene glycol, diethylene glycol, propylene glycol, dipropylene glycol, 1,4-butanediol, 1,6-hexanediol, neopentyl glycol, glycerol and trimethylolpropane, and pentaerythritol. The polyol used is preferably chosen from among the group defined previously for preparation of the polyester polyol of the invention, consisting of monoethylene glycol, diethylene glycol, butanediol, and their mixtures. Among the dicarboxylic acids that can advantageously be used, let us mention, as examples, succinic acid (C4), glutaric acid (C5), adipic acid (C6), and their mixtures. The aliphatic dicarboxylic acid used is preferably adipic acid.

In order to prepare the prepolymers by known means, the diisocyanate(s) employed are heated in a reaction vessel to a temperature of between 40 and 80°C, then the polyester polyol resin(s) is (are) added with agitation. The temperature rises to between 80 and 110°C and this temperature is maintained for 1 to 6 hours. The content of NCO groups is checked and adjusted to a value of between 2 and 25% by weight.

10

As an example, the existing polyether polyols that can be used in this invention to prepare the prepolymer include those described in patent EP 0,582,385 obtained by polymerization of ethylene oxide with another cyclic oxide such as propylene oxide, butylene oxide, or tetrahydrofuran in the presence of a polyfunctional initiator such as water or polyols, for example monoethylene glycol, diethylene glycol, propylene glycol, dipropylene glycol, cyclohexane dimethanol, resorcinol, bisphenol A, glycerol, trimethylolpropane, 1,2,6-hexanetriol, or pentaerythritol. The prepolymer is prepared by known means through a reaction of the polyisocyanates with the polyether polyol resin or resins at a temperature of between 50 and 90°C for a reaction time of between 0.5 and 4 hours.

20

15

25

The urethane elastomer, also referred to herein as polyurethane foam in the case of expanded urethane elastomer, obtained in the presence of a foaming agent, can then be prepared by any conventional process known to specialists in the field, through a reaction of the prepolymer and the hydroxylated compound in the presence of a catalyst and possibly in the presence of a foaming agent, a cross-linking agent, or a chain-lengthening agent, and/or other known additives. The elastomer is advantageously prepared by causing two components—one component consisting of the prepolymer and a second so-called hydroxylated component formed of a mixture of the hydroxylated compound consisting of at least one polyester polyol of the invention prepared as indicated above—with the catalyst and optionally the foaming agent and the additives.

5

The preparation of polyurethane foams is described in greater detail, specifically in the Polyurethane Handbook, G. Oertel, Hanser Publishers, 2nd Edition, 1993.

The catalysts are advantageously amine-type catalysts such as 1,4-diaza-[2,2,2]bicyclo octane (also called triethylenediamine) as is or locked in the carboxylic acids, and possibly organic salts of tin or bismuth.

10

For the foaming agent, one can use water possibly mixed with a surfactant and/or organic compounds having a low melting point.

15

20

25

The optional cross-linking and chain-lengthening agents are compounds having at least two hydrogen atoms likely to react with the isocyanates and having a molecular weight of less than 500. These compounds are, for example, compounds having hydroxyl and/or amino and/or carboxyl groups such as monoethylene glycol, 1,4butanediol, trimethylolpropane or glycerol, methylenebis-(2-chloroaniline), methylenebis-(dipropylaniline), diethyl-toluenediamine, trimethylene glycol di-paminobenzoate, isophorone diamine, triethanolamine, and triisopropanolamine, which may possibly be alkoxylated.

The surfactants used are advantageously of the wax and mineral oil type, which are mixtures of hydrocarbons having a naphthenic or paraffinic structure, or of the silicone oil type optionally modified by alkoxylation.

Other additives may also be introduced by known means, such as emulsifying agents, foam stabilization agents, fire-retardants, pigments, dyes, fillers, resistivity reduction agents (anti-static agents), anti-aging stabilizers, or substances having an antifungal or anti-bacterial action.

The prepolymer and hydroxylated compound are each loaded into a tank of the machine and each is raised to a temperature of between 20 and 60°C. The flows of both the prepolymer and the hydroxylated compound that must enter into the reaction are

5

10

15

20

25

adjusted so as to obtain in the machine's mixing head the desired NCO:OH molar ratio close to the stoichiometry between the NCO groups of the prepolymer and the reactive groups of the hydroxylated compound with respect to the NCOs. For reasons of implementation owing to the machines, the NCO:OH ratio by weight is generally between 0.5 and 1.5, preferably between 0.6 and 1.2. The mixture is introduced into a mold raised to a temperature of between 20 and 60°C. It is either injected into the closed mold or poured into the open mold, in which case the mold is immediately closed.

The polyurethane foams of this invention can be used to manufacture soles consisting of a single layer, referred to as monodensity, or so-called "combined" soles consisting of two layers: a relatively compact wear layer having a density of about 0.9 to 1.25, and a comfort layer having a lower density of about 0.3 to 0.5. The layer of monodensity soles and the comfort layer of combined soles are made of expanded urethane elastomer, whereas the wear layer may be made of expanded or non-expanded urethane elastomer.

## **EXAMPLES**

The examples given below, solely as non-limiting illustrations, will provide a better understanding of the invention. In the examples the polyester polyol of this invention is used solely on the hydroxylated compound side, with the polyurethane foams being made from known prepolymer compositions. Unless indicated otherwise, the ratios specified in the examples below are expressed in moles.

A. Preparation of Polyester Polyol Resins

# 1. Polyester polyol resin A1:

Resin A1, having an average molecular weight of 2000 by weight, was obtained from the following ingredients:

5

- 50:50 molar mixture of phthalic anhydride/sebacic acid
- diethylene glycol (DEG)
- catalyst: FOMREZ SUL11A

10

15

20

25

In an esterification reactor equipped with a fractionating column are introduced the sebacic acid, phthalic anhydride, and DEG. We heat with agitation up to 220-230°C in a nitrogen atmosphere in the presence of a catalyst consisting of 84 ppm of FOMREZ SUL11A marketed by the Crompton company (this catalyst is a tin derivative). The reaction is continued for 24 hours. The properties of the resulting A1 resin, that is, the appearance at 25°C, the hydroxyl number (OHN) in mg of KOH per gram, the acid number (AN) in mg of KOH per gram, the percentage of water by weight, the Hazen color, and the viscosity at 25°C, are indicated in Table I below:

# TABLE I

Properties	Resin A1
Appearance at 25°C	Clear liquid
OHN (mg KOH/g)	56.1
AN (mg KOH/g)	0.2
Water (%)	0.01
Color (Hazen)	150
Viscosity at 25°C	25,800
(mPa/s)	

# 2. Polyester polyol resins A2-A4

Resins A2 to A4 were prepared in the same way as resin A1 using the same catalyst and the ingredients indicated in Table II below:

30

# **TABLE II**

5

10

15

20

	Resin A2	Resin A3	Resin A4
Acid	PA:SA	PA:SA	PA:DA
	30:70	60:40	60:40
Polyol	DEG	DEG	DEG
	100	100	100

PA: phthalic anhydride; SA: sebacic acid (C10); DA: dodecanedioic acid (C12); DEG: diethylene glycol

The hydroxyl number (OHN), acid number (AN), and viscosity at 25°C of resins A2-A4 are indicated in Table III below:

**TABLE III** 

Properties	Resin A2	Resin A3	Resin A4
OHN (mg KOH/g)	56.2	55.8	54.2
AN (mg KOH/g)	0.2	0.2	0.4
Viscosity at 25°C (mPa/s)	10,600	47,400	42,300

3. Comparative polyester polyol resins (not part of the invention)

Comparative resins B1 through B4 were made according to the process described previously for resin A1 using the same catalyst and the ingredients indicated in Table IV below:

**TABLE IV** 

25		Resin B1	Resin B2	Resin B3	Resin B4	Resin B5
	Acid	AA 100	AA 100	AA/SA 50:50	SA 100	AA 100
	Polyol	MEG:BD 70:30	MEG:DEG:TM P 59:39:2	MEG:BD 70:30	DEG 100	MEG:DEG:TM P 59:39:2

AA: adipic acid; PA: phthalic anhydride; SA: sebacic acid; MEG: monoethylene glycol; DEG: diethylene glycol; BD: 1,4-butanediol; TMP: trimethylol propane

The hydroxyl number (OHN), acid number (AN), and viscosity at 25, 50 or

60°C of resins B1-B5 are indicated in Table V below:

10

5

15

20

25

30

35

TABLE V

Properties	Resin B1	Resin B2	Resin B3	Resin B4	Resin B5
OHN (mg KOH/g)	56.8	55.9	54.1	54.8	38.5
AN (mg KOH/g)	0.4	0.5	0.2	0.2	0.6
Viscosity (mPa/s) at °C	2000/50	11,500/25	1000/60	780/60	29,500/25

# B. Prepolymers:

Two types of prepolymers (P1 and P2) of known compositions were used. These two prepolymers were prepared from the ingredients indicated in Table VI below:

TABLE VI

	Prepolymer P1	Prepolymer P2
NCO%	16.5	19
Isocyanate(s)	pure 4,4'-MDI 100	pure 4,4'-MDI:modified MDI* 80:20 by weight
Polyester	Resin B2	Resin B5

<sup>\*</sup> Modified MDI: Desmodur CD by Bayer

# C. Preparation of Polyurethane Foams

For the production of polyurethane foams, we cause a prepolymer as defined in section B and a hydroxylated component consisting of a polyester polyol resin as defined in section A, to react in a mixture of water, catalyst, and chain-lengthening

5

10

15

20

25

agent, and possibly an antistatic agent (the product marketed by the Great Lakes company under the name "Catafor F"), and/or a surfactant (the product marketed by the "Crompton Corporation" under the name "NIAX Silicone SR 393") at a temperature of between 20 and 60°C. The quantities of prepolymer and hydroxylated component that must react with each other are adjusted starting with the NCO:OH stoichiometric ratio. The NCO:OH ratios actually used are given, by weight, in the result tables below.

In order to illustrate the improvement in hydrolysis resistance achieved by the process of this invention, in examples 1-5 we have compared elastomers of the invention, made using resin A1, with known elastomers made using one of comparative resins B1 through B5. Example 6 concerns elastomers of the invention manufactured using resins A2-A4 of the invention.

For each batch of urethane elastomer obtained, we conduct (in accordance with ISO standard 5423: 1992, Appendix E) a hydrolysis test on several test pieces from the batch, a test in which each test piece is placed inside an enclosure for several days at 70°C and 100% relative humidity (saturating water vapors). For each batch, we measure the breaking load (according to DIN standard 53504) on the test pieces taken before hydrolysis and on several test pieces later taken after the hydrolysis tests conducted as indicated above on various days, and then we calculate the average breaking load and the corresponding retention percentages. We also measure the following mechanical properties for each batch:

- Shore A hardness according to DIN standard 53505;
- elongation according to DIN standard 53504;
- resistance to tearing according to DIN standard 53515;
- resistance to flexing according to the Ross Flex method with starter cut

5

10

15

# EXAMPLE 1:

This example involves injection molding of monodensity sole layers on a Desma machine performed with elastomers E1, E2, and E3 of the invention, prepared using resin A1 in the hydroxylated component, whose formulation is indicated below, and prepolymer P1 in ratios by weight of NCO:OH of 98:100, 100:100, and 102:100, respectively, and comparative elastomers C1, C2, and C3 prepared from the same prepolymer P1 and resin B1 in place of resin A1, in the same formulation of hydroxylated component and in the same identical ratios. The reaction parameters and results of the physical and chemical property tests are given in Table VII below.

Formulation of the hydroxylated component (% by weight):

**TABLE VII** 

	C1-C3	E1-E3	
Resin B1 Resin A1 MEG Trimethylol propane Triethylene diamine Niax Silicone SR 393 Water	90.25% 7.87% 0.5% 0.6% 0.4% 0.38%	90.25% 7.87% 0.5% 0.6% 0.4% 0.38%	

20

**TABLE VII** 

Track free time (s) 5-6 5-6 5-6  Tack free time (s) 21-23 18-20  Setting time (s) 34-36 32-34  Free density 0.34 0.33  Lifting time in min 2 2 2  Molding temperature in 55 55  Physical and Mechanical Properties*  Molding density 0.6 0.6  Shore A hardness 49 50 51 43 42  Elongation % 520 475 505 540 535  Breaking load  before hydrolysis MFa 7.4 7.0 6.7 6.0 6.1  Bydrolysis at 9 days MFa (% ret.)  Bydrolysis at 12 days MFa (% ret.)  Bydrolysis at 15 days MFa (% ret.)  Bydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2  Bydrolysis at 22 days MFa (% ret.)  Bydrolysis at 22 days MFa (% ret.)  Bydrolysis at 25 days MFa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the							
Propolymer   B1		Compara	tive elastom	ers C1-3	Elastomers E1-3		
Reaction Farameters		C1	C2	сз	E1	E2	E3
NCO:OE ratio   98:100   100:100   102:100   98:100   100:100   1	Prepolymer				P1		
NCO:OH ratio 98:100 100:100 102:100 98:100 100:100 10  Creme time (s) 5-6 5-6  Tack free time (s) 21-23 18-20  Setting time (s) 34-36 32-34  Free density 0.34 0.33  Lifting time in min 2 2 2  Molding temperature in 55 55  Fhysical and Mechanical Properties*  Molding density 0.6 0.6  Shore A hardness 49 50 51 43 42  Elongation % 520 475 505 540 535  Breaking load  before hydrolysis MPa 7.4 7.0 6.7 6.0 6.1  Bydrolysis at 9 days MPa (4.5 (61) 5.5 (79) 54. (81) 5.5 (92) 5.7 (95) 5.8 (8 ret.)  Bydrolysis at 12 days MPa (4.6 (47) 5.1 (85) 5.2 (97) MPa (8 ret.)  Bydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2 (97) MPa (8 ret.)  Bydrolysis at 22 days 0 0 0 0 2.9 (48) 4.15 (69) 4.6 (77) Fearing N/ma (8 ret.)  Bydrolysis at 25 days MPa (6 ret.)  Bydrolysis at 25 days 0 0 0 0 2.9 (48) 4.15 (69) 4.6 (77) MPa (8 ret.)  Bydrolysis at 25 days 0 0 0 0 2.9 (48) 4.15 (69) 4.6 (77) MPa (8 ret.)  Bydrolysis at 25 days MPa (8 ret.)  Bydrolysis at 25 days 0 0 0 0 2.9 (48) 4.15 (69) 4.6 (77) MPa (8 ret.)  Bydrolysis at 25 days MPa (8 ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the	Polyester polyol resin		B1			A1	
Tack free time (s) 5-6 5-6  Tack free time (s) 21-23 18-20  Setting time (s) 34-36 32-34  Free density 0.34 0.33  Lifting time in min 2 2 2  Molding temperature in 55 55  Physical and Mechanical Properties*  Molding density 0.6 0.6  Shore A hardness 49 50 51 43 42  Elongation % 520 475 505 540 535  Breaking load  before hydrolysis MFa 7.4 7.0 6.7 6.0 6.1  Hydrolysis at 9 days MFa (% ret.)  Bydrolysis at 12 days MFa (% ret.)  Bydrolysis at 15 days MFa (% ret.)  Bydrolysis at 18 days MFa (% ret.)  Bydrolysis at 18 days MFa (% ret.)  Bydrolysis at 2 days MFa (% ret.)  Bydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2  Bydrolysis at 25 days MFa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the			Reaction I	Parameters			
Tack free time (s)   21-23   18-20	NCO:OH ratio	98:100	100:100	102:100	98:100	100:100	102:100
Setting time (a)         34-36         32-34           Free density         0.34         0.33           Lifting time in min         2         2           Molding temperature in 'C'         55         55           Physical and Mechanical Properties*           Molding density         0.6         0.6           Shore A hardness         49         50         51         43         42           Elongation %         520         475         505         540         535           Breaking load         520         475         505         540         535           Bydrolysis at 9 days MPa (** ret.)         4.5 (61)         5.5 (79)         54. (81)         5.5 (92)         5.7 (95)         5.8           Bydrolysis at 12 days MFa (** ret.)         3.6 (49)         4.8 (69)         5.0 (75)         5.1 (85)         5.4 (90)         5.2           Bydrolysis at 15 days MFa (** ret.)         1.0 (15)         1.7 (24)         2.0 (30)         4.9 (81)         5.4 (90)         5.2           Bydrolysis at 18 days MFa (* ret.)         0         0         4.0 (67)         5.1 (85)         5.2           Bydrolysis at 22 days MFa (* ret.)         0         0         0         4.6 (77)         5.1 (85)<	Creme time (s)	<u></u>	5-6			5-6	
### Description of the image of	Tack free time (s)		21-23		<u> </u>	18-20	
Lifting time in min   2	Setting time (s)		34-36			32-34	
Molding temperature in 55	Free density		0.34			0.33	
Physical and Mechanical Properties*  Molding density 0.6 0.6  Shore A hardness 49 50 51 43 42  Elongation % 520 475 505 540 535  Breaking load  Defore hydrolysis MPa 7.4 7.0 6.7 6.0 6.1  Bydrolysis at 9 days MPa 4.5 (61) 5.5 (79) 54. (81) 5.5 (92) 5.7 (95) 5.6 (% ret.)  Hydrolysis at 12 days 3.6 (49) 4.8 (69) 5.0 (75) 5.1 (85) 5.4 (90) 5.2 MPa (% ret.)  Bydrolysis at 18 days 1.0 (15) 1.7 (24) 2.0 (30) 4.9 (81) 5.4 (90) 5.2 MPa (% ret.)  Bydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2 MPa (% ret.)  Bydrolysis at 22 days 0 0 0 4.0 (67) 4.8 (80) 4.8 MPa (% ret.)  Bydrolysis at 25 days 0 0 0 2.9 (48) 4.15 (69) 4.8 MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the	Lifting time in min	<u> </u>	2		Í	2	
Molding density 0.6 0.6  Shore A hardness 49 50 51 43 42  Elongation % 520 475 505 540 535  Breaking load  before hydrolysis MPa 7.4 7.0 6.7 6.0 6.1  Hydrolysis at 9 days MPa 4.5 (61) 5.5 (79) 54. (81) 5.5 (92) 5.7 (95) 5.8 (% ret.)  Hydrolysis at 12 days MPa (3.6 (49) 4.8 (69) 5.0 (75) 5.1 (85) 5.4 (90) 5.8 MPa (8 ret.)  Hydrolysis at 15 days 1.0 (15) 1.7 (24) 2.0 (30) 4.9 (81) 5.4 (90) 5.2 MPa (% ret.)  Hydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2 MPa (% ret.)  Hydrolysis at 22 days MPa (% ret.)  Hydrolysis at 25 days 0 0 0 4.0 (67) 4.8 (80) 4.8 MPa (% ret.)  Hydrolysis at 25 days 0 0 0 2.9 (48) 4.15 (69) 4.4 MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the			55			55	
Shore A hardness 49 50 51 43 42  Elongation % 520 475 505 540 535  Breaking load  before hydrolysis MPa 7.4 7.0 6.7 6.0 6.1  Hydrolysis at 9 days MPa (% ret.)  Hydrolysis at 12 days 3.6 (49) 4.8 (69) 5.0 (75) 5.1 (85) 5.4 (90) 5.5 MPa (% ret.)  Hydrolysis at 15 days 1.0 (15) 1.7 (24) 2.0 (30) 4.9 (81) 5.4 (90) 5.2 MPa (% ret.)  Hydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2 MPa (% ret.)  Hydrolysis at 22 days 0 0 0 4.0 (67) 4.8 (80) 4.6 MPa (% ret.)  Hydrolysis at 25 days 0 0 0 2.9 (48) 4.15 (69) 4.6 MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the		Physic	cal and Mecha	nical Prope	rties*		
Elongation % 520 475 505 540 535  Breaking load  before hydrolysis MPa 7.4 7.0 6.7 6.0 6.1  Hydrolysis at 9 days MPa 4.5 (61) 5.5 (79) 54. (81) 5.5 (92) 5.7 (95) 5.8 (% ret.)  Hydrolysis at 12 days 3.6 (49) 4.8 (69) 5.0 (75) 5.1 (85) 5.4 (90) 5.5 MPa (% ret.)  Hydrolysis at 15 days 1.0 (15) 1.7 (24) 2.0 (30) 4.9 (81) 5.4 (90) 5.2 MPa (% ret.)  Hydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2 MPa (% ret.)  Hydrolysis at 22 days 0 0 0 4.0 (67) 4.8 (80) 4.8 MPa (% ret.)  Hydrolysis at 25 days 0 0 0 2.9 (48) 4.15 (69) 4.4 MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the	Molding density	<u> </u>	0.6			0.6	<b> </b>
Breaking load         before hydrolysis MPa       7.4       7.0       6.7       6.0       6.1         Hydrolysis at 9 days MPa (% ret.)       4.5 (61)       5.5 (79)       54. (81)       5.5 (92)       5.7 (95)       5.8 (92)         Hydrolysis at 12 days MPa (% ret.)       3.6 (49)       4.8 (69)       5.0 (75)       5.1 (85)       5.4 (90)       5.5 (92)         Hydrolysis at 15 days MPa (% ret.)       1.0 (15)       1.7 (24)       2.0 (30)       4.9 (81)       5.4 (90)       5.2 (90)         Hydrolysis at 18 days MPa (% ret.)       0.4 (5)       1.3 (19)       1.8 (27)       4.6 (77)       5.1 (85)       5.2 (85)         Hydrolysis at 22 days MPa (% ret.)       0       0       0       4.0 (67)       4.8 (80)       4.6 (80)         Hydrolysis at 25 days MPa (% ret.)       0       0       0       2.9 (48)       4.15 (69)       4.4 (69)         Tearing N/mm       30       28       27       26       24         Ross Flex at 20°C, kilocycles for 500% propagation of the       >200       180       170       >200       >200	Shore A hardness	49	50	51	43	42	44
before hydrolysis MPa 7.4 7.0 6.7 6.0 6.1  Hydrolysis at 9 days MPa 4.5 (61) 5.5 (79) 54. (81) 5.5 (92) 5.7 (95) 5.8 (% ret.)  Hydrolysis at 12 days 3.6 (49) 4.8 (69) 5.0 (75) 5.1 (85) 5.4 (90) 5.5 MPa (% ret.)  Hydrolysis at 15 days 1.0 (15) 1.7 (24) 2.0 (30) 4.9 (81) 5.4 (90) 5.2 MPa (% ret.)  Hydrolysis at 18 days 0.4 (5) 1.3 (19) 1.8 (27) 4.6 (77) 5.1 (85) 5.2 MPa (% ret.)  Hydrolysis at 22 days 0 0 0 4.0 (67) 4.8 (80) 4.8 MPa (% ret.)  Hydrolysis at 25 days 0 0 0 2.9 (48) 4.15 (69) 4.4 MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C, kilocycles for 500% propagation of the	Elongation %	520	475	505	540	535	495
Hydrolysis at 9 days MPa	Breaking load				<del>,</del>		
(% ret.)       Hydrolysis at 12 days       3.6 (49)       4.8 (69)       5.0 (75)       5.1 (85)       5.4 (90)       5.8         MPa (% ret.)       Hydrolysis at 15 days       1.0 (15)       1.7 (24)       2.0 (30)       4.9 (81)       5.4 (90)       5.2         MPa (% ret.)       Hydrolysis at 18 days       0.4 (5)       1.3 (19)       1.8 (27)       4.6 (77)       5.1 (85)       5.2         MPa (% ret.)       0       0       0       4.0 (67)       4.8 (80)       4.8         Hydrolysis at 22 days       0       0       0       4.0 (67)       4.8 (80)       4.8         MPa (% ret.)       0       0       0       2.9 (48)       4.15 (69)       4.4         Tearing N/mm       30       28       27       26       24         Ross Flex at 20°C, kilocycles for 500% propagation of the       >200       180       170       >200       >200	before hydrolysis MPa	7.4	7.0	6.7	6.0	6.1	6.1
MPa (* ret.)         Hydrolysis at 15 days       1.0 (15)       1.7 (24)       2.0 (30)       4.9 (81)       5.4 (90)       5.2         MPa (* ret.)       1.3 (19)       1.8 (27)       4.6 (77)       5.1 (85)       5.2         MPa (* ret.)       0       0       0       4.0 (67)       4.8 (80)       4.8         Hydrolysis at 22 days MPa (* ret.)       0       0       0       2.9 (48)       4.15 (69)       4.4         MPa (* ret.)       30       28       27       26       24         Ross Flex at 20°C, kilocycles for 500% propagation of the       >200       180       170       >200       >200		4.5 (61)	5.5 (79)	54. (81)	5.5 (92)	5.7 (95)	5.8 (97)
MPa (% ret.)         Bydrolysis at 18 days       0.4 (5)       1.3 (19)       1.8 (27)       4.6 (77)       5.1 (85)       5.2         MPa (% ret.)       0       0       0       4.0 (67)       4.8 (80)       4.8         MPa (% ret.)       0       0       0       2.9 (48)       4.15 (69)       4.6         MPa (% ret.)       0       0       0       2.9 (48)       4.15 (69)       4.6         MPa (% ret.)       30       28       27       26       24         Ross Flex at 20°C, kilocycles for 500% propagation of the       >200       180       170       >200       >200		3.6 (49)	4.8 (69)	5.0 (75)	5.1 (85)	5.4 (90)	5.5 (92)
MPa (% ret.)  Hydrolysis at 22 days  MPa (% ret.)  Hydrolysis at 25 days  MPa (% ret.)  Tearing N/mm  30  28  27  26  24  Ross Flex at 20°C, kilocycles for 500% propagation of the		1.0 (15)	1.7 (24)	2.0 (30)	4.9 (81)	5.4 (90)	5.2 (87)
MPa (% ret.)  Hydrolysis at 25 days 0 0 0 2.9 (48) 4.15 (69) 4.4  MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C,		0.4 (5)	1.3 (19)	1.8 (27)	4.6 (77)	5.1 (85)	5.2 (87)
MPa (% ret.)  Tearing N/mm 30 28 27 26 24  Ross Flex at 20°C,		o	o	o	4.0 (67)	4.8 (80)	4.8 (80)
Ross Flex at 20°C, >200 180 170 >200 >200 kilocycles for 500% propagation of the		0	0	0	2.9 (48)	4.15 (69)	4.4 (73)
kilocycles for 500% propagation of the	Tearing N/mm	30	28	27	26	24	23
<del></del>	kilocycles for 500%	>200	180	170	>200	>200	120

<sup>\*</sup> performed on sheets 6 mm thick

Elastomers E1-E3 of the invention are made under implementation conditions identical to those of elastomers C1-C3.

5

10

15

20

In this example the breaking load was measured after 9, 12, 15, 18, 22, and 25 days under hydrolysis conditions. Figure 1A and 1B show, in the form of graphs, the hydrolysis resistance results of comparative elastomers C1-C3 and elastomers E1-E3 of the invention, respectively, which are indicated in Table VII. For a better graphical comparison, the hydrolysis resistance results of elastomer E2 and comparative elastomer C2 are shown graphically in Figure 1C. Elastomers E2 and C2 correspond to the elastomers yielding the best results in the hot indentation test as performed on the foam during industrial molding.

These results clearly show that the elastomers of the invention have considerably greater hydrolysis resistance than that of the comparative elastomers, and this in spite of a slightly lower Shore A hardness in this particular case.

# **EXAMPLE 2**

This example involves the molding of wear layers for combined soles made with elastomer E4 of the invention prepared using resin A1 in the hydroxylated component, and comparative elastomer C4 made with resin B2, with both elastomers, E4 and C4, being prepared with the same prepolymer P1. In this example the formation of the hydroxylated compound based on resin A1 was enriched with MEG so as to obtain elastomers providing comparable Shore A hardness values. The reaction parameters and results are given in Table VIII below. Formulation of the hydroxylated component for C4 and E4 (% by weight):

25

	C4	E4	
Resin A1 Resin B2 MEG Triethylene diamine Water	94.5% 4.6% 0.8% 0.1%	93.5% - 5.6% 0.8% 0.1%	

TABLE VIII

	Comparative elastomer C4	Elastomer E4
Prepolymer	P1	
Polyester polyol resin	B2	A1
	Reaction Parameters	
NCO:OH ratio	63:100	73:100
Creme time (s)		
Tack free time (s)	16-18	19-21
Setting time (s)	26-28	36-38
Lifting time in min	2	2
Molding temperature in °C	45	45
	Physical and Mechanical Properties*	
Molding density	1.05	<u> </u>
Shore A hardness	60-61	63-64
Elongation %	565	575
Breaking load		
before hydrolysis MPa	12.1	13.7
Hydrolysis at 5 days MPa (% ret.)	69	79
Hydrolysis at 7 days MPa (% ret.)	43	72
Hydrolysis at 9 days MPa (% ret.)	32	70
Hydrolysis at 14 days MPa (% ret.)	0	47
Hydrolysis at 21 days MPa (% ret.)	0	32
Tearing N/mm	53	44
Ross Flex at 20°C, kilocycles for 500% propagation of the starter cut	65	70

<sup>\*</sup> performed on sheets 6 mm thick

The hydrolysis resistance results of comparative elastomer C4 and elastomer E4 of the invention indicated in Table VIII are shown graphically in Figure 2. Elastomer E4 has a markedly greater hydrolysis resistance than that of elastomer C4 while being easy to implement and retaining comparable or better physical properties.

5

# **EXAMPLE 3:**

This example involves the molding of layers for monodensity soles made from elastomer E5 of the invention, prepared using resin A1 in the hydroxylated component and prepolymer P2, and comparative elastomer C5 made with resin B1 and prepolymer P1. In this example the formulations were adjusted with MEG so as to obtain elastomers having comparable Shore A hardness values. The reaction parameters and results are given in Table IX below.

Formulation of the hydroxylated component for C5 and E5 (% by weight):

15

10

	C5	E5
Resin A1 Resin B1 MEG Triethylene diamine Trimethylol propane Niax Silicone SR 393 Water	- 90.6% 7.5% 0.6% 0.5% 0.4% 0.4%	89.1% - 9% 0.6% 0.5% 0.4% 0.4%

25

5 TABLE IX

	Comparative elastomer C5	Elastomer E5
Prepolymer	P1	P2
Polyester polyol resin	B1	A1
	Reaction Parameters	<del></del>
NCO:OH ratio	99:100	95:100
Creme time (s)	5-6	4-5
Tack free time (s)	23-25	16-18
Setting time (s)	43-45	32-34
Free density	0.32	0.30
Lifting time in min	2.25	2
Molding temperature in °C	45	45
	Physical and Mechanical Properties*	<del></del>
Molding density	0.6	0.6
Shore A hardness	52	54
Elongation %	380	310
Breaking load		
before hydrolysis MPa	8.5	7.9
Hydrolysis at 10 days MPa (% ret.)	6.5 (76)	5.9 (75)
Hydrolysis at 15 days MPa (% ret.)	3.4 (40)	4.9 (62)
Hydrolysis at 20 days MPa (% ret.)	1.4 (16)	4.1 (52)
Hydrolysis at 25 days MPa (% ret.)	0	2.3 (32)
Hydrolysis at 30 days MPa (% ret.)	0	1.1 (14)
Tearing N/mm	33	30
Ross Flex at 20°C, kilocycles for 100% propagation of the starter cut	>150	>150

<sup>\*</sup> performed on sheets 6 mm thick

Elastomer E5 of the invention has a flexing resistance value at least equal to that of comparative elastomer C5.

The hydrolysis resistance results of the comparative elastomer and the elastomer of the invention indicated in Table IX are shown graphically in Figure 3. These results show that elastomer E5 of the invention also has excellent hydrolysis resistance properties compared to elastomer C5.

5

10

15

20

25

# **EXAMPLE 4**

This example involves the molding of comfort layers for combined soles made from elastomers E6-E9 of the invention, prepared using resin A1 in the hydroxylated component and prepolymer P1, and comparative elastomers C6-9 made with resin B1 and prepolymer P1. In this example the formulations were adjusted with MEG so as to obtain elastomers having comparable Shore A hardness values. The reaction parameters and results are given in Table X below.

Formulation of the hydroxylated component for C5 and E5 (% by weight):

	C6-9	E6-9
Resin A1 Resin B1 MEG Catafor F Triethylene diamine (TEDA) Trimethylol propane Niax Silicone SR 393 Water	- 86.77% 7.2% 4.0% 0.6% 0.5% 0.4% 0.53%	85.47% - 8.5% 4.0% 0.6% 0.5% 0.4% 0.53%

0201-PC

116:100 日9 Elastomers E6-9 35-40 17-21 0.24 2-6 45 71 A1 N 114:1 日8 00 112: 100 E7 Reaction Parameters 110 :10 **E**6 TABLE X 105:1 00 Comparative elastomers C6-9 80 103: 100 C8 23-26 40-44 0.24 4-5 45 P1 B1 N 101:1 C799: C6 Polyester polyol resin Setting time (s) Lifting time in min Molding temperature in Tack free time (s) Creme time (s) NCO:OH ratio Free density Prepolymer

20

21

		Phys	Physical a	and Mechanical	anical	Properties*	rties*	
Molding density		)	0.5					0.5
Shore A hardness	39	39	62	39	40	40	40	4.0
Elongation %	490	470	044	425	435	415	380	375
Breaking load								
before hydrolysis MPa	5.2	5.4	5.5	5.6	5.0	5.3	5.6	5.6
Hydrolysis at 10 days MPa (% ret.)	3.4 (65)	4.3 (80)	5 (91)	5.2 (93)	4 (80)	4.6 (87)	5.4 (100)	5.9 (100
Hydrolysis at 15 days MPa (% ret.)	1.8 (35)	2 (37)	2 (36)	2.4 (43)	4 (80)	4.5 (85)	5 (94)	5.2 (93)
Hydrolysis at 20 days MPa (% ret.)	0.8 (15)	0.9 (17)	1.1 (20)	1.1 (20)	2.8 (56)	3.1 (58)	4.2 (79)	4.6 (82)
Hydrolysis at 25 days MPa (% ret.)	0	0	0	0	1.3 (26)	1.5 (28)	2 (38)	2.4 (43)
Hydrolysis at 30 days MPa (% ret.)	0	0	0	0	1.1 (22)	1.4 (26)	2 (38)	1.5 (27)
Tearing N/mm	23	21	21	21	23	22	22	21

\* performed on sheets 6 mm thick

Elastomers E6-E9 of the invention demonstrated a very good ability to be implemented under customary conditions and showed physical properties at least equal to those of comparative elastomers C6-C9.

10

Figures 4A and 4B show in graphic form the hydrolysis resistance results of the comparative elastomers and the elastomers of the invention, respectively, indicated in Table X. For a better graphical representation, the hydrolysis resistance results of elastomer E8 and comparative elastomer C8 are shown simultaneously in Figure 4C. Elastomers E8 and C8 correspond to the elastomers having the best results in the indentation test. These results show that elastomers E6-9 of the invention also possess excellent properties of hydrolysis resistance.

15

20

In addition, elastomer E8 of the invention and comparative elastomer C8 were subjected to an impact absorption test. The test consisted of a falling weight test in which an 8.5 kg striker is dropped onto the sample from a height of 50 mm. These conditions are designed to simulate the impact at a man's heel while running. The striker is circular with a semi-convex side having a curvature radius of 37.5 mm. An accelerometer and a displacement sensor attached to the striker make it possible to measure the maximum deceleration (m/s2) of the striker upon impact with the sample, the returned energy in % (the quantity of energy returned after the impact determined from the rebound height of the striker), and the maximum penetration in mm (the maximum ram compression of the sample upon impact). The results are given in Table XI below:

5

TABLE XI

	Elastomer C8	Elastomer E8
Thickness (mm)	6	6
Maximum deceleration (m/s2)	445	350
Maximum penetration (mm)	3	3
Energy returned (%)	24	15

10

The deceleration and returned energy values, which represent the impact absorption properties, show that elastomer E8 of the invention has markedly better impact absorption properties than those of comparative elastomer C8.

15

# **EXAMPLE 5**

This example involves moldings of wear layers for combined soles made with elastomer E10 of the invention, prepared with resin A1, and with comparative elastomers C10, C11, and C12 prepared with resins B1, B3, and B4, respectively. The formulations of the hydroxylated component for E10, C10, C11, and C12 are comparable to those described previously in example 2 (resin + MEG + triethylene diamine + water). The reaction parameters and results are given in Table XII below:

ស

10

15

20

25

TABLE XII

	Comparative elastomer C10	Comparative elastomer C11	Comparative elastomer C12	Elastomer E10
Prepolymer	P1	P1	P1	PJ
Polyester polyol resin	B1	ВЗ	B4	A1
		Reaction Parameters		
NCO:OH ratio	99:100	68:100	68:100	65:100
Creme time (s)	7-9	8-10	10-12	6-8
Tack free time (s)	31-33	29-31	28-30	21-23
Setting time (s)	56-58	53-55	48-50	42-45
Free density	0.7	0.75	72.0	0.7
Lifting time in min	2.75	2.75	2.5	2.5
Molding temperature in °C	20	05	50	50
	Phy	Physical and Mechanical Properties *		
Molding density	1			
Shore A hardness	61-63	63-65	60-61	60-62
Elongation in %	530	530	800	540
Brenking load				
before hydrolysis MPa	18.7	21.3	16.3	15.4
Hydrolysis at 7 days, % retention	43	53	72	62
Hydrolysis at 14 days, % retention	8	23	51	64
Hydrolysis at 21 days, % retention	0	2	14	35
Tearing N/mm	58	55	45	96
Ross Flex at 20°C, kilocycles for 500%	09	>200	130	>200

\* performed on sheets 6 mm thick

5

These results show that simply adding sebacic acid (elastomer C12) is not enough to achieve the hydrolysis resistance obtained with the polyesters of this invention.

# **EXAMPLE 6**

10

This example involves moldings of wear layers for combined soles made with elastomers E11-E13 of the invention prepared using resins A2, A3, and A4. The formulations of the hydroxylated component for E10, C10, C11, and C12 are similar to those described previously in example 2 (resin + MEG + triethylene diamine + water). The reaction parameters and results are given in Table XIII below.

## **TABLE XIII**

15

20 25

30

35

40

	Elastomer El I	Elastomer E12	Elastomer E13
Prepolymer	P1	PI	Pl
Polyester polyol resin	A2	A3	A4
	Reaction Parameters		<b>_</b>
NCO:OH ratio	70:100	70:100	70:100
Creme time (s)	8-10	7-9	8-10
Tack free time (s)	28-30	20-22	24-26
Setting time (s)	48-50	42-44	48-50
Free density	0.77	0.76	0.72
Lifting time in min	2.5	2.25	2.25
Molding temperature in °C	50	50	50
	Physical and Mechanical Prop	erties *	
Molding density	1	1	11
Shore A hardness	61-62	59-61	60-62
Elongation in %	490	515	490
Breaking load			
before hydrolysis MPa	14.2	15.2	13.8
Hydrolysis at 7 days, % retention	70	68	69
Hydrolysis at 14 days, % retention	60	56	63
Hydrolysis at 21 days, % retention	33	40	42
Tearing N/mm		55	58
Ross Flex at 20°C, kilocycles for 500% propagation of starter cut	60	70	75

<sup>\*</sup> performed on sheets 6 mm thick

Elastomers E11-E13 made from resins A2-A4 have very good resistance to hydrolysis. Compared to Tables VIII and XII, these results show that the combination

5

of the acids in this invention provides a marked improvement in the resistance to hydrolysis compared to the previous technology.

The combined results of examples 1 through 5 demonstrate that the process of this invention makes it possible to achieve a substantial systematic improvement in the hydrolysis resistance of urethane elastomers.